

# High refractive index low-loss aluminium oxide waveguides

(Student Paper)

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## ABSTRACT

A high refractive index ( $n \sim 1.715$  at 633 nm of wavelength) aluminium oxide slab waveguide has been developed, which exhibits very low losses from the near-UV to the near infrared wavelength range. Slab waveguide losses as low as 1.8 dB/cm at 407 nm and less than 0.1 dB/cm at 1550 nm of wavelength have been experimentally characterized. Such low losses are limited by surface scattering. The layer was deposited by reactive sputter coating with an aluminium target, a set substrate temperature of 700 C and an oxidation state of the target of 5 %. The very high optical quality of this material, in combination with rare-earth ion doping, could pave the way towards high-gain on-chip amplifiers in different wavelength ranges, on-chip lasers and non-linear applications.

**Keywords:** Aluminium oxide, refractive index, low loss, sputtering.

## 1. INTRODUCTION

The excellent optical properties of aluminium oxide ( $\text{Al}_2\text{O}_3$ ) make it a very interesting material for integrated photonics[1]–[4]. The large transparency window from 150 nm up to 5500 nm allows for applications from the UV to the mid-infrared [5]. Rare earth ion doped amplifiers and lasers have been shown in amorphous aluminium oxide with wavelengths ranging from 880 nm to up to 2  $\mu\text{m}$  [4], [6]–[8]. Integration with passive photonic platforms has been the subject of much research efforts recently [7], [9]–[12].

Reactive sputter deposition can be utilized to deposit high optical quality thin films for integrated optical circuits with reasonably high deposition rates. The potential of reactive sputter coating of optical grade amorphous  $\text{Al}_2\text{O}_3$ , has previously been demonstrated by Wörhoff et al. [13], where they optimized the sputter coating process to achieve a propagation loss of 0.3 and 0.1 dB/cm at a wavelength of 633 nm and 1550 nm respectively for a low refractive index material ( $n \sim 1.65$  at 633 nm). Low losses were achieved by optimizing the process over a large parameter space. Substrate temperature was reported as the main parameter that influences propagation loss. They also compared DC and RF sputter coating, the latter delivering higher layer quality, most likely due to reduced arcing during the process.

Similar propagation losses were obtained by Magden et al. [14] for an optimized process with the addition of a bias voltage to the substrate. The deposited material also exhibited a low refractive index ( $n \sim 1.578$  at 1550 nm).

In this work, an  $\text{Al}_2\text{O}_3$  layer with high refractive index (i.e.,  $n \sim 1.715$  at 633 nm) is deposited by increasing the set temperature of the substrate during sputtering to 700 °C. Very low slab losses of  $\sim 1.8$  dB/cm at 407 nm and well below 0.1 dB/cm at 1550 nm were experimentally demonstrated in this high refractive index  $\text{Al}_2\text{O}_3$  layer.

## 2. DEPOSITION AND CHARACTERIZATION OF ALUMINIUM OXIDE THIN FILMS

The  $\text{Al}_2\text{O}_3$  layers were deposited using an AJA ATC 15000 RF reactive co-sputtering system [13] on 10 cm diameter silicon wafers with 8  $\mu\text{m}$  thick thermal oxide. The system can host three 2 inch RF guns, one of which was installed with an aluminium target (Al, 99.9995 % purity). The main deposition chamber is evacuated to a base pressure of 0.1  $\mu\text{Torr}$  (i.e., 13  $\mu\text{Pa}$ ) to prevent incorporation of hydroxide ions in the layer, which induce absorption losses around 750 nm, 970 nm, and 1400 nm [15], and when the layer is doped with rare-earth ions, induce luminescence quenching [16]. The depositions are performed with a constant RF power of 200 W applied to the aluminium target. Before the start of the process, the bias voltage of the target as a function of oxygen flow into the chamber is measured. For low oxygen flows, the target is still mostly metallic. As the oxygen flow increases, the target starts oxidizing. Since the secondary electron emission yield of  $\text{Al}_2\text{O}_3$  is higher than that of aluminium, the voltage drops to a minimum value that corresponds to a fully oxidized target [17]–[19]. A relatively low sputter flux combined with a large pump flux ensure that the bias curve evolves smoothly for our process[20]. The oxidation state of the target can be defined as

$$\eta_{ox} \equiv \frac{V_m - V_{O_2}}{V_m - V_{ox}} \quad (1)$$

where  $V_m$  and  $V_{O_2}$  are the cathode bias voltage for a fully metallic target and a fully oxidized target surface respectively and  $V_{ox}$  is the bias voltage corresponding to the oxygen flow giving a relative oxidation state,  $\eta_{ox}$ . The relative oxidation state is used to compensate for the target erosion, which results in a changing bias curve,

changing deposition rate and a change in the ad-particle energy. Using the oxidation state of the target method allows us to reproduce layers over most of the target lifetime. In this work, an oxidation state of the target of 5% was selected, since this value resulted in low loss amorphous low refractive index ( $n \sim 1.65$ )  $\text{Al}_2\text{O}_3$  when the depositions were carried out at a set substrate temperature of 580 °C. All parameters used for the deposition of the high index layer at 700 °C are given in Table 1.

Table 1 Optimized Rf reactive sputtering process parameters

Parameter	Value	Unit
Argon flow	30	sccm
Oxygen flow	1.1	sccm
Relative bias voltage	5	%
Pressure	3.4	mTorr
Substrate set temperature	700	°C
Target-substrate distance	15.2	cm
Power	200	W

The refractive index, thickness and propagation loss of the layer were measured using a Metricon 2010/M prism coupler instrument with a propagation loss module. The measurements were carried out at 407 nm, 474 nm, 532 nm, 632.8 nm, 827 nm, 983 nm, 1312 nm and 1550 nm [21]. The refractive index results are presented in Figure 1A, showing a refractive index increase of circa 0.05 compared to those obtained by Wörhoff et al. [13]. The increased index of refraction likely indicates an increased density of the layer. An increased layer density can reduce scattering from micro-voids and other growth defects in the layer due to a lower chance of occurrence as a result of the increased ad-particle mobility [22], [23]. Figure 1B shows the slab propagation loss as a function of wavelength. As it can be seen, a very low propagation loss of 0.025 dB/cm at 1550 nm was achieved, which is below the resolution of the measuring system. Propagation losses of 1.8 dB/cm at 407 nm are similar to the losses recently reported for low refractive index  $\text{Al}_2\text{O}_3$  deposited by atomic layer deposition (ALD) [24]. The traced line in Figure 1B represents the Rayleigh scattering relation, given by the following equation,

$$\alpha(\lambda) = \alpha_0 \left( \frac{\lambda_0}{\lambda} \right)^4 \quad (2)$$

where  $\alpha_0$  indicates the measured propagation loss at  $\lambda_0$ . The loss value at 632.8 nm was used to create the given curve. The excellent correspondence of the line trace suggests that the main contributor to the increased losses in the visible and near-UV is volume and surface scattering in the layer. Atomic force microscopy measurements show a surface roughness of the layer of 2.4 nm rms, which corresponds to the value calculated from the measured losses, which gives an expected surface roughness of  $2.8 \pm 0.7$  nm rms[25]

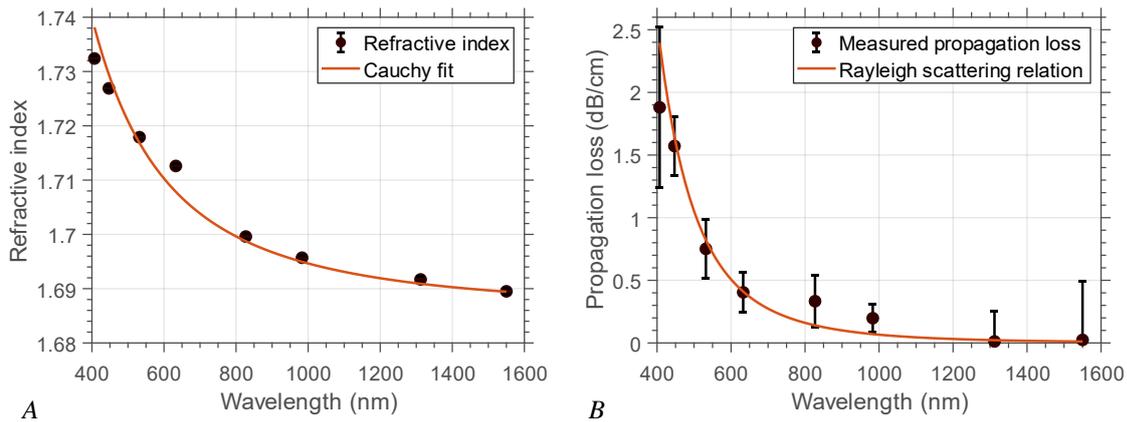


Figure 1 (A) Refractive index values (error  $\pm 0.0005$ ), with a Cauchy fit (coefficients:  $B=1.6852$  and  $C=0.00899 \mu\text{m}^2$ ) (B) Propagation loss as a function of wavelength, fitted with the Rayleigh scattering relation indicating the loss mechanism of the layer is scattering dominated.

### 3. CONCLUSIONS

The development of high refractive index  $\text{Al}_2\text{O}_3$  layer with low propagation loss from the near-UV to the mid-IR is very promising for integrated photonics in wavelength ranges where other materials do not perform. The increased refractive index of the layer allows for increased mode confinement. Finally, as the density increase is caused by a substrate temperature increase. It is possible that the layers exhibit a polycrystalline structure corresponding to a layer morphology of zone T or 2 [22], which needs to be further investigated. Polycrystallinity

might allow for increased rare earth ion dopant concentrations, where concentration quenching is prevented by a crystal lattice enforced inter-atomic distance of the dopant.

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