## High-gain waveguide amplifiers based on potassium rare-earth double tungstates

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The down-scaling of the rare-earth-doped amplifier to chip-scale device length is beneficial for signal amplification within short-reach interconnects such as optical backplanes [1, 2] and for the realization of novel active/passive integrated optical chips. Thus far, with the demonstration of 20 dB net gain [3], high bit-rate operation at 170 Gb/s [4], and gain bandwidth over 80 nm [5] in the C-band, as well as 14.4 dB net gain at 1064 nm wavelength [6], rare-earth-doped amorphous  $Al_2O_3$  is a very promising material for waveguide amplifiers. The corresponding peak gain per unit length of  $Er^{3+}$ -and  $Nd^{3+}$ -doped  $Al_2O_3$  are  $\sim$ 2.0 dB/cm [5] and 6.3 dB/cm [6], respectively. To further increase the gain per unit length, a host material with higher transition cross-sections and higher active-ion solubility is needed. In this work, it is shown that highly  $Yb^{3+}$ -doped and  $Er^{3+}$ -doped monoclinic potassium rare-earth double tungstate,  $KRE(WO_4)_2$  epitaxial layers match these two conditions and produce higher gain per unit length.

Yb³+-doped and Er³+-doped KRE(WO<sub>4</sub>)<sub>2</sub> layers were grown on undoped potassium yttrium double tungstate, KY(WO<sub>4</sub>)<sub>2</sub>, substrates with the dimension of 1 mm × 10 mm × 10 mm by liquid phase epitaxy [7]. Optically inert Lu³+ and Gd³+ are included in the epitaxial layer for: i) maintaining lattice matching conditions with respect to the substrate, ii) increasing the refractive index contrast between active layer and substrate, and iii) maximizing the active ion concentration. The index of refraction of the substrate at  $E||n_m|$  is ~2.0 for the wavelength range of 1000-1550 nm and the refractive index contrast between the active layer and the substrate is ~0.012.

Figure 1(a) shows the absorption and emission cross-sections of the  $KYb_{0.57}Gd_{0.43}(WO_4)_2$  layer determined with the procedure described in [8]. The 57 at.%  $Yb^{3+}$  in the epitaxial layer corresponds to an active ion density of  $\sim 3.8 \times 10^{21}$  cm<sup>-3</sup>. The peak absorption and emission cross-sections are  $13.1 \times 10^{-20}$  cm<sup>2</sup> and  $16.2 \times 10^{-20}$  cm<sup>2</sup>, respectively, at  $\sim 981$  nm, which is about an order of magnitude higher than those of  $Al_2O_3$ :Yb<sup>3+</sup> [9]. The transition cross-sections of  $Er^{3+}$ -doped  $KRE(WO_4)_2$  layers are displayed in Fig. 1(b). The luminescence spectra and luminescence lifetime measurements were performed on a layer with the lowest available  $Er^{3+}$  concentration of 0.75 at.% ( $\sim 0.47 \times 10^{20}$  cm<sup>-3</sup>) to avoid reabsorption effects, while the reciprocity theorem was used to calculate the absorption cross-section. The  $Er^{3+}$  peak transition

cross-sections are  $\sim 4.3 \times$  higher in potassium double tungstates than in Al<sub>2</sub>O<sub>3</sub>.

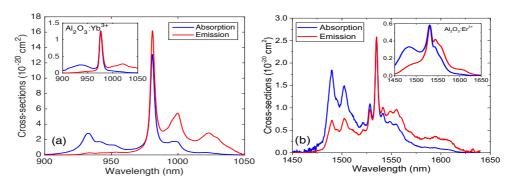


Fig. 1. Transition cross-sections of (a) Yb $^{3+}$ -doped and (b) Er $^{3+}$ -doped potassium rare-earth double tungstate epitaxial layers. The insets show the corresponding transition cross-sections for the same active ion doped into Al $_2$ O $_3$  [5, 9] for comparison purpose.

The luminescence lifetime measured from KYb $_{0.57}$ Gd $_{0.43}$ (WO $_4$ ) $_2$  is merely ~7% less than the lifetime from low-doped (1.2 at.% Yb $^{3+}$ ) sample (245  $\mu$ s), indicating a very weak lifetime quenching. The averaged lifetime of the first excited state ( $^4I_{13/2}$ ) for the samples with 0.75-10 at.% Er $^{3+}$  is ~3.05 ms. No sign of lifetime quenching was observed for the samples within the range of Er $^{3+}$  concentrations investigated. However, like in many Er $^{3+}$ -doped materials, excitation-density-dependent energy-transfer upconversion significantly shortens the luminescence-decay time.

Pump-probe measurements are conducted to determine the optical gain. By launching both optical beams perpendicular to the 32 µm-thick  $KYb_{0.57}Gd_{0.43}(WO_4)_2$  layer, a net gain of >900 dB/cm is determined at the 981 nm signal wavelength by pumping at 932 nm wavelength. The reduction of transition cross-sections at elevated temperatures [8] due to pump-induced heating may be the gain-limiting mechanism. The 6 at.%  $Er^{3+}$ -doped  $KRE(WO_4)_2$  channel waveguides, micro-structured using ion-beam etching, produce gain >10 dB/cm at the 1534.75 nm signal wavelength by pumping at 984.5 nm, which is much higher than most of the singly  $Er^{3+}$ -doped materials. The gain of the  $Er^{3+}$ -doped  $KRE(WO_4)_2$  waveguides is limited by energy-transfer upconversion and excited-state absorption processes. As a result, higher  $Er^{3+}$  concentration does not lead to better gain.

In summary, Yb³+- or Er³+-doped KRE(WO $_4$ ) $_2$  epitaxial layers exhibit large transition cross-sections and limited luminescence lifetime quenching at high doping concentrations. High optical gain of >900 dB/cm for highly Yb³+-doped layers (57 at.%) at 981 nm wavelength and >10 dB/cm for the Er³+-doped layers (6 at.%) at 1534.75 nm wavelength were demonstrated.

## References

- $[1] \quad \text{R. Dangel et al., Opt. Express $\textbf{23}$, 4736-4750, 2015}.$
- [2] F. E. Doany et al., IEEE Trans. Adv. Packag. 32, 345-359, 2009.
- [3] S. A. Vázquez-Córdova et al., Opt. Express 22, 25993-26004, 2014.
- [4] J. D. B. Bradley et al., Opt. Express **24**, 22201-22208, 2009.
- [5] J. D. B. Bradley et al., J. Opt. Soc. Am. B 27, 187-196, 2010.
- [6] J. Yang et al., Appl. Phys. B 101, 119-127, 2010.
- [7] S. Aravazhi et al., Appl. Phys. B **111**, 433-446, 2013.
- [8] Y. S. Yong et al., Opt. Express **24**, 26825-26837, 2016.
- [9] L. Agazzi, Ph.D. thesis, University of Twente, Enschede, The Netherlands, p. 35, 2012.